

# Domain-structure analysis in structural phase transitions



## 2022 Spring Festival Crystallographic School and Workshop on Crystal-field Applications

University of Science and Technology, Beijing, China,  
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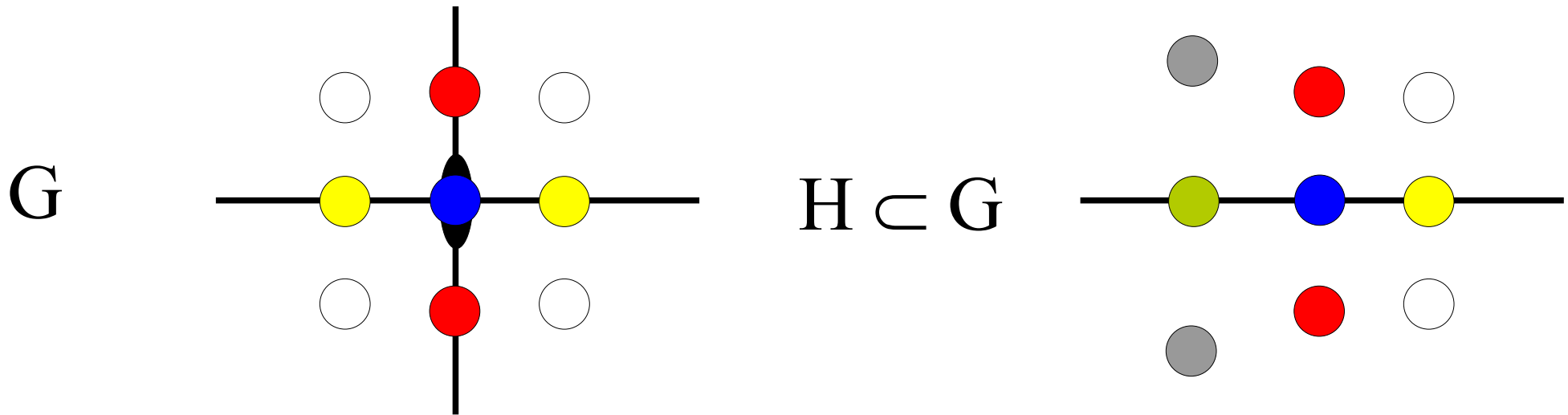
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











# Classification of subgroups

- $G$  itself and  $\{1|000\}$  are the two **trivial subgroups** of  $G$ .
- All the other subgroups of  $G$  are called **proper subgroups**.
- If  $H$  is a proper subgroup of  $G$  ( $G \supset H$ ) such that there is no intermediate group between  $G$  and  $H$ , then  $H$  is called a **maximal subgroup** of  $G$ .
- If  $G$  and  $H$  have the same translation subgroup, *i.e.* the same lattice (warning: not just the same *type* of lattice!), then  $H$  is called a **t-subgroup (translationengleiche subgroup)** of  $G$ .
- If  $G$  and  $H$  have the same type of point group (*i.e.* they belong to the same geometric crystal class), then  $H$  is called a **k-subgroup (klassengleiche subgroup)** of  $G$ .
- If  $H$  is a  $k$ -subgroup of  $G$  and is of the same type as  $G$  (same Hermann-Mauguin symbol), then  $H$  is called an **i-subgroup (isomorphic subgroup)** of  $G$ .

# Effect of the symmetry reduction of Wyckoff positions



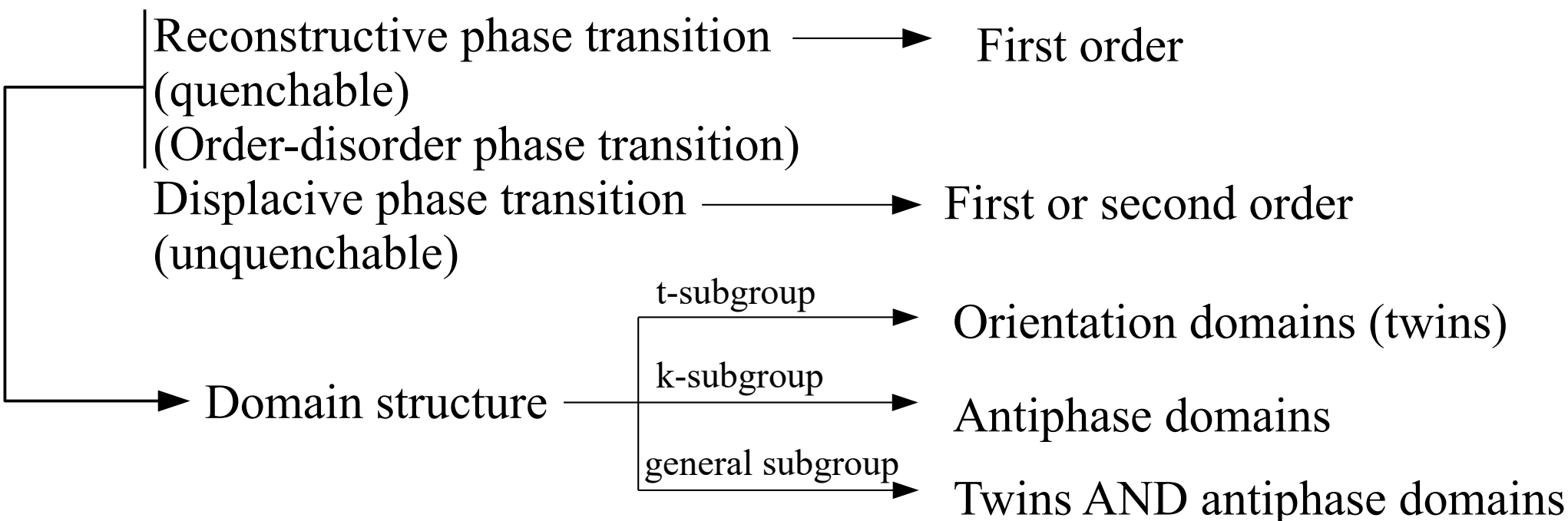
	Multiplicity	Site-symmetry group (s.s.g)		Multiplicity	Site-symmetry group (s.s.g)	
	4	1	<b>Splitting</b>	 	2	1
	2	$m..$		 	1	$m..$
	2	$.m.$	<b>Reduction of s.s.g</b>		2	1
	1	$mm2$			1	$m..$

# Symmetry reduction following a phase transition

Thermodynamic classification of phase transitions (Ehrenfest and Tisza)

First order, second order, lambda...

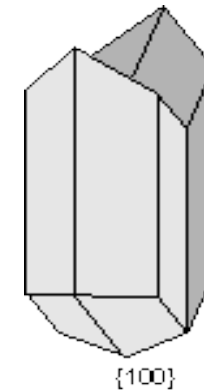
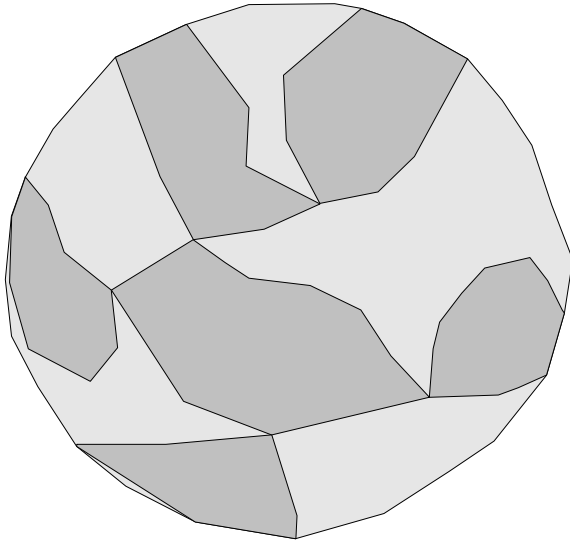
Mechanistic classification of phase transitions (Buerger)



**Index of subgroup: number of domain states**

**Cosets: Orientation (twin) and position (antiphase) of the domains**

# Domains vs. domain states



Two twinned individuals

Two **domain states** (variants)

dark grey: 6 **domains**

light grey: 5 **domains**

# Example

$\text{SrTiO}_3$  perovskite structure type, space-group type  $Pm\bar{3}m$

At low T (below 105 K) the relative rotation of the octahedra results in a phase transition

$$Pm\bar{3}m \rightarrow I4/mcm$$

$$c \rightarrow 2c$$

$$V \rightarrow 4V \text{ (V is the volume of the unit cell)}$$

Determine the consequences of this phase transition, namely:

The number of domain states

The classification of the domain states (twin, antiphase)

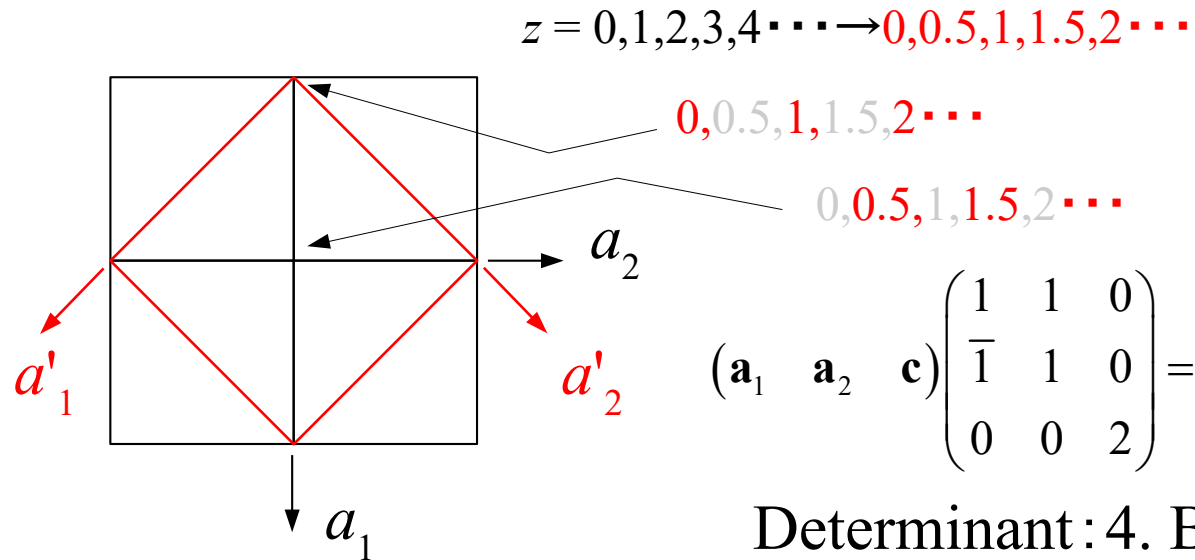
The (representative) operation mapping two domain states

# Example (solution)

$$Pm\bar{3}m \rightarrow I4/mcm$$

$$c \rightarrow 2c$$

$$V \rightarrow 4V$$



$$(\mathbf{a}_1 \quad \mathbf{a}_2 \quad \mathbf{c}) \begin{pmatrix} 1 & 1 & 0 \\ \bar{1} & 1 & 0 \\ 0 & 0 & 2 \end{pmatrix} = (\mathbf{a}'_1 \quad \mathbf{a}'_2 \quad \mathbf{c}')$$

Determinant: 4. BUT  $P \rightarrow I$

$$i_L = 2$$

$$i_P = |m\bar{3}m|/|4/mmm| = 48/16 = 3$$

$$i = i_P \cdot i_L = 3 \cdot 2 = 6$$

6 domain states

- 3 orientation domain states.  
Representative operation:  $3^+_{[111]}$  and  $3^-_{[111]}$
- For each of the three, 2 antiphase domain states.  
Representative operation:  $t(\frac{1}{2}\frac{1}{2}0)$

# Effect of translation of Fourier Transform

$$\rho_T(\mathbf{r}) = \rho(\mathbf{r}-\mathbf{t}) \quad \mathbf{t} = \text{translation vector} \quad F_T(\mathbf{r}^*) = \mathcal{T}^{-1}[\rho_T(\mathbf{r})]$$

$$\begin{aligned} F_T(\mathbf{r}^*) &= \int \rho(\mathbf{r}-\mathbf{t}) \exp(2\pi i \mathbf{r}^* \cdot \mathbf{r}) dV_r = \\ &= \exp(2\pi i \mathbf{r}^* \cdot \mathbf{t}) \int \rho(\mathbf{r}-\mathbf{t}) \exp(2\pi i \mathbf{r}^* \cdot \mathbf{r} - \mathbf{t}) dV_r = \\ &= F(\mathbf{r}^*) \exp(2\pi i \mathbf{r}^* \cdot \mathbf{t}) \end{aligned}$$

Translation of  $\rho(\mathbf{r})$  by a vector  $\mathbf{t}$  in direct space is equivalent to modifying the Fourier transform by the phase factor  $\exp(2\pi i \mathbf{r}^* \cdot \mathbf{t})$  in reciprocal space, without change in the modulus  $|F(\mathbf{r}^*)|$ , but the real and imaginary part of  $F(\mathbf{r}^*)$  are multiplied by  $\cos(2\pi \mathbf{r}^* \cdot \mathbf{t})$  and  $\sin(2\pi \mathbf{r}^* \cdot \mathbf{t})$  respectively. A description of a transform is thus *origin dependent*.



# Diffraction pattern of a domain structure

$$F_T(\mathbf{r}^*) = F(\mathbf{r}^*) \exp(2\pi i \mathbf{r}^* \cdot \mathbf{t})$$

$$I(hkl) \propto |F(hkl)|^2$$

G group of the parent phase



H group of the daughter phase

*translationengleiche* subgroup  
(same lattice, lower point group)

**Twin domains**  
(differing by orientation)

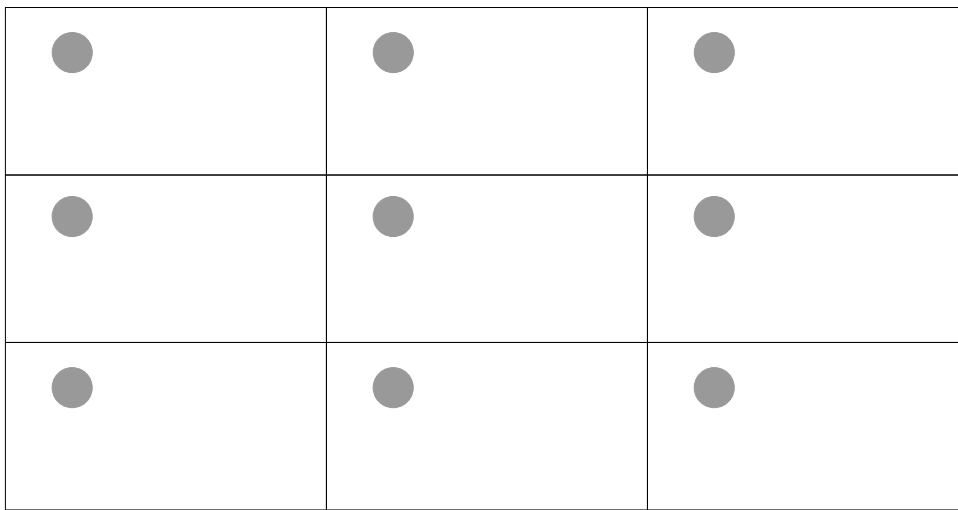
Visible in the diffraction pattern, which appears as the overlap of the diffraction pattern of each domain state.

*klassengleiche* subgroup  
(sublattice, same point group)

**Antiphase domains**  
(differing by position)

Invisible in the diffraction pattern  
(effect on the *phase*)  
Visible by electron microscopy.

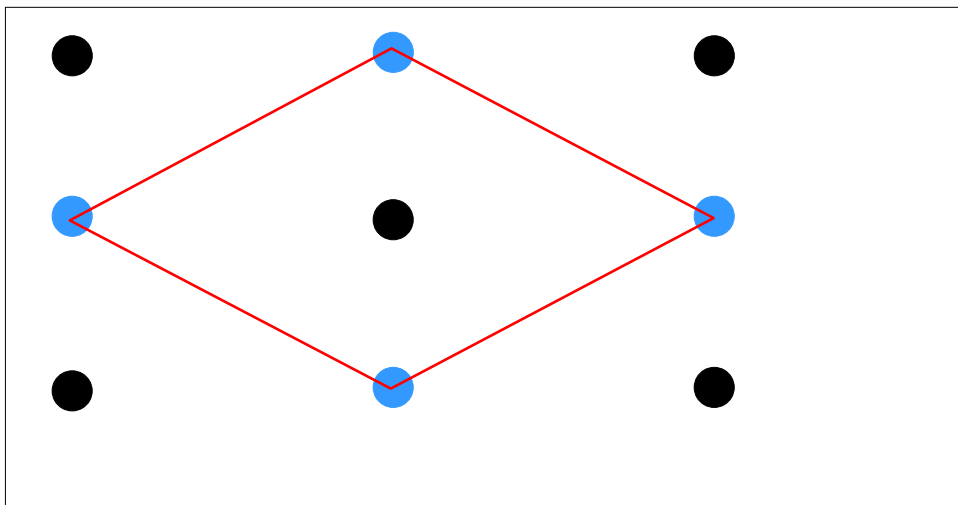
# Superlattice vs. sublattice



Basic structure (aristotype)  
 Space-group  $G$   
 Translation subgroup  $T(G)$

↓ **supercell**  
 $T(H)$  is a subgroup of  $T(G)$   
 → **sublattice**

↑ **subcell**  
 $T(G)$  is a supergroup of  $T(H)$   
 → **superlattice**



Derivative structure (hettotype)  
 $H \subset G$   
 $T(H) \subset T(G)$